Finite element simulation of cold pilgering of ODS tubes

Esteban Vanegas Márqueza, Katia Mocellin, Louise Toualbi, Yann de Carlan, Roland E. Logé

HAL Id: hal-00759474

https://minesparis-psl.hal.science/hal-00759474

Submitted on 30 Nov 2012

where only monotonic laws are used to simulate in 3D the pilgering process. The full stroke numerical simulation of the cold pilgering process leads to the determination of the strain path undegone by a material point. This task in new comparing to litterature.

2. Simulation set up

Optimizing numerically the whole process with a long tube is simply not feasible at the time being, because of the huge computation time involved. Consequently, a *short tube* approach is used. A pileging pass where deformation is close to 25% is simulated. The tube is turned 39° and moved forward (feed) 1.7 mm after each stroke. A material point takes 120 strokes to pass through the working zone. Initial position of the simulated systems is displayed in Figure 1. Lagrangian numerical sensors have been placed at the mid-length of the tube at different angles.

Figure 1. FEM simulation set-up.

3. Monotonic constitutive model

A standard constitutive model adapted to described monotonic loadings, tensile is used. The elastic-visco-plastic Hansel-Spittel law is chosen, it is given by :

$$\sigma_0(\bar{\varepsilon}) = A e^{m_1 T} T^{m_9} \bar{\varepsilon}^{m_2} e^{m_4 \bar{\varepsilon}} \left(1 + \bar{\varepsilon}\right)^{m_5 T} e^{m_7 \bar{\varepsilon}} \dot{\bar{\varepsilon}}^{m_3} \dot{\bar{\varepsilon}}^{m_8 T}$$

where $\bar{\varepsilon}$ is the equivalent deformation (total strain), $\dot{\bar{\varepsilon}}$ the equivalent deformation rate (total strain rate), T is the temperature and A, m_{1-9} are the regression coefficients. A isothermal tensile test is considered , no strain rate effects is assumed: parameters m_3 and m_8 are therefore set to zero.

4. Cyclic elastic–plastic constitutive model

The cyclic deformation induced by cold pilgering is fairly well approximated by a sequence of compression tests

along two perpendicular directions [4]. The constitutive model considered in this study is therefore based on Chaboche for the cyclic behavior, it is denote as *cyclic C/C*. The latter utilizes multi-components forms of kinematic and isotropic hardening variables with non-linear evolutionary rules to describe the hysteresis loops and the transient behavior of the material.

The formulation chosen here is the following [5]. The elastic domain is defined by:

$$f = J_2\left(\underline{\sigma} - \sum_i \underline{X}_i\right) - \sum_i R_i - \sigma_y \leq 0$$

In the previous equation σ_y is the initial yield surface size. $\underline{X}$ is the kinematic internal stress tensor and R is the isotropic hardening. The formulation of the kinematic hardening tensor is based on the non-linear Chaboche kinematic model.

5. Damage

The criterion introduced by Latham and Cockroft and used in this work establishes the damage function from the principal stresses. This damage criterion was already used for VMR cold pilgering (e.g. [6]) and seems to be a good indicator to predict defects frequency. Compressive stresses have a negligible effect on damage compared with tensile ones; so in its original version this criterion does not take them into account. The classical damage function used in [6] is the sum of the product of the maximum tensile stress at each stroke. This function represents an indicator of the material state.

Considering the HPTR cold pilgering process, and the experimental evidence of longs cracks along the rolling axis, it is proposed to consider each strain increments in all directions ($\Delta\varepsilon_i$, $i = rr, \theta\theta, zz, r\theta, \theta z, rz$). Similarly, the expression of the maximal stress is modified in order to consider only positive stress components. Finally, authors have shown that damage occurs for mechanical states where the triaxiality is higher than -1/3. A modified cumulative damage function (Lat&Co) is then proposed:

$$Lat\&Co_{(i)} = \sum_{\substack{Stroke \\ triaxiality \geq -1/3}} \sigma_i^+ * \Delta\varepsilon_i^+$$

Where $i = rr, \theta\theta, zz, r\theta, \theta z, rz$, $\sigma_i^+ = \max(0, \sigma_i)$, $triaxiality = \sigma_H/\sigma_{eq}$ with σ_H the hydrostatic pressure and σ_{eq} the equivalent stress. Thus, it is possible to obtain a cumulative damage field in each mechanical solicitation direction.

Longitudinal damage function is maximal when comparing with other directions (not showed here). Figure 2 shows the sensor (located at mi-length and mid-thickness). Longitudinal damage function evolution for each constitutive law. Longitudinal damage is highest for the

cyclic C/C behavior law. Even with the huge stress values predicted by monotonic laws; the longitudinal damage value is lower than the cyclic law. Hence, when the cyclic constitutive law is taken into account, the material follows more *damaging* situations.
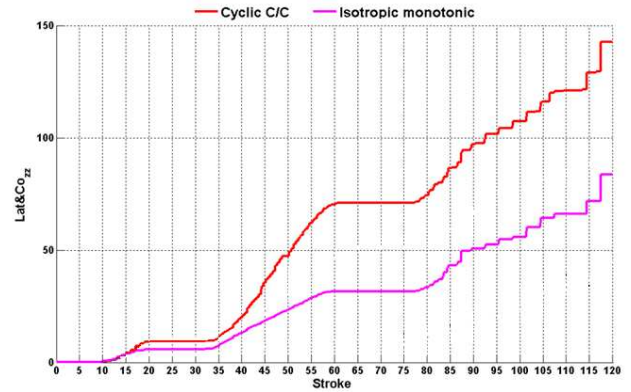

Figure 2. Longitudinal damage function history in sensor S2 (in MPa).

Looking at Figure 2 and considering the *cyclic C/C* behavior law, damage function increases dramatically between strokes 35-55 and 80-110 for the longitudinal component. Monotonic behavior law also leads to an increase, but with lower intensity.

6. Conclusions

Accurate modeling of the material constitutive behavior is one of the most important features needed in order to simulate the process properly. Full numerical simulations considering the monotonic and cyclic constitutive behavior of ODS steels were carried out. A systematic analysis of all strokes has helped defining which stroke could contribute mainly to the oligocyclic fatigue of the material. Future work will include a numerical simulation of ODS tubes cold pilgering considering anisotropic-cyclic constitutive behavior. The influence of this feature on the mechanical history undergone by a material point during the process will be assessed.

7. References

[1] Y. de Carlan *et al.*, J. Nucl. Mater. 386-388 (2009) 430-432.
[2] L. Toualbi *et al.*, Proceedings of DIANA I workshop ''Dispersion Strengthened steels for advanced nuclear applications'', Aussois (France) April 4-8, 2011, J. Nucl. Mater.
[3] P. Montmitonnet *et al.*, J. Mater. Proc. Tech. 125-126 (2002) 814-820.
[4] E. Vanegas-Marquez *et al.*, J. Nucl. Mater. 420 (2012) 479-490.
[5] J.-L. Chaboche, Int. J. Sol. Struc. 34 (1997) 2239-2354.
[6] E. Girard *et al.*, J. Nucl. Mater. 294 (2001) 330-338.