Dual cells with mixed protonic-anionic conductivity for reversible SOFC/SOEC operation

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Abstract

The dual cell concept is a novel design for solid oxide fuel cells operating at intermediate temperature. The cell comprises a series of five layers with different compositions, alternating two dense electrolytes and three porous layers, i.e. the outer electrodes and a central membrane. The dual cell concept makes it possible to separate the compartment for water formation from both fuel and oxidant chambers. Such a three-chamber configuration gives many advantages related to fuel dilution, materials corrosion, and reversibility between fuel cell and electrolyser operational modes (SOFC/SOEC) at high temperature. Dual conductivity (protonic/anionic) can be achieved by joining two dense BaCe0.85Y0.15O3–G (BCY) and Ce0.85Y0.15O2–G (YDC) electrolytes through a porous ceramic central membrane made up of both materials. Complete anode-supported dual cells have been fabricated by a combination of pressing, casting, printing, wet spraying, and plasma spraying techniques. Electrochemical tests carried out by impedance spectroscopy showed the feasibility of the concept and successful reversible operation of the dual cell. The fabrication route, the microstructural and electrochemical testing results are reported in this work, and partially compared to simulated results from an electrochemical model developed describing the dual cell concept.

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1. Introduction

Solid oxide fuel cell (SOFC) technology is receiving increasing scientific and industrial interest for a number of stationary and portable applications, like combined heat and power (CHP) systems and auxiliary power units (APUs) [1]. The same technology is being considered for the reverse operation, i.e. the production of $\text{H}_2$ gas by high-temperature electrolysis (solid oxide electrolysis cell, SOEC), especially combined with renewable sources or nuclear plants [2].

The main limitations to SOFC/SOEC systems are represented by stack degradation over long-term operation, and resistance to thermal and redox cycling, which could be mitigated by lowering the operating temperature, e.g. by adopting proton conducting electrolytes [3]. The presence of water at either cathode or anode is also unfavourable because of negative effects on fuel utilisation, electrode stability, and metallic interconnect durability [4].

The architecture of the Dual Membrane Cell (DM-Cell) combines all the advantages of protonic SOFCs without the drawbacks associated with the presence of water at the electrodes. In such a three-chamber cell water is produced in a mixed $\text{H}^+$ and $\text{O}^{2-}$ conducting membrane, which also allows easy reverse operation as an electrolysis cell without any adjustment to the composition of feed gases.

Fig. 1 shows the schematic of a DM-Cell principle. When operating in SOFC mode, $\text{H}^+$ and $\text{O}^{2-}$ ions are formed at the anode and cathode, respectively, migrate through the protonic and anionic electrolyte towards the dual membrane placed in the middle of the cell, and there recombine to form water vapour. The SOEC operating mode is obtained by just reversing the polarisation and feeding pure water vapour in the Dual Membrane compartment. The formation of $\text{H}_2\text{O}$ in a separate compartment should reduce corrosion of interconnects and remove the need for drying in the case of fuel gas recirculation.

![Fig. 1](image-url)
2. Proof of the concept and modelling

The concept of a DM-Cell was proven in a series of dedicated experiments focusing on four different criteria: (1) establishment of a stable OCV, (2) measurement of electric current under load (i-V curve), (3) detection of water formation by electrochemical impedance spectroscopy (EIS), and (4) correlation between polarisation and water production in the DM compartment [5]. Fig. 2 clearly shows the humidity change in the N₂ gas streaming through the DM compartment, across polarisation switching cycles in a complete DM-Cell (criterion 4).

The operation of the DM fuel cell was simulated by means of a microkinetic mathematical model which takes into account charge transfer, mass transfer, and electrochemical reactions in the cell, and enables the prediction of the steady-state and dynamic response of the DM-Cell under varying operating conditions [6].

The model was validated on polarisation data experiments carried out for the proof of concept (Fig. 3a). The effect of the DM thickness on the cell performance was investigated. A thick layer hosts a large number of active reaction sites, with reduced current flux per single contact point. As a consequence, thick layers show low activation resistance, but they may be affected by high ohmic losses. Moreover, thin layers have low ohmic losses, but due to the reduced number of active reaction sites, they are characterised by high activation losses. As a result, for any given selection of materials, morphology (particle size \( d_p = 0.4 \) µm, volume fraction of pore \( \varepsilon = 0.25 \)) and operating conditions (i.e. temperature), an optimum thickness can be identified (Fig. 3), with balanced ohmic and activation resistances to provide minimum energy losses. The calculated polarisation curve reveals that 120 µm corresponds to the thickness leading to the best cell performance, which gives comfortable latitude for shaping (Fig. 3b).
3. Experimental

The feasibility and the potential performance of the DM-Cell have been explored by applying easily scaled-up fabrication processes, like tape casting and spraying. BaCe$_{0.85}$Y$_{0.15}$O$_3$–G (BCY) powders were employed for the fabrication of protonic electrolyte and cermet anode layers, Ce$_{0.85}$Y$_{0.15}$O$_2$–G (YDC) powders were adopted for the anionic electrolyte. A mixture (50 vol%) was used to realise DM layers. The average particle size for both compositions was about 200 nm, and for plasma spraying an additional granulation step was carried out in order to form larger aggregates. Cathodic layers were made of La$_{0.6}$Sr$_{0.4}$Co$_{0.2}$Fe$_{0.8}$O$_3$–G (LSCF). In the case of tape cast cells, all layers of the DM-Cell were made of BCY, in order to take advantage of the dual (anionic-protonic) conductivity of this perovskite structure [7] and simplify the multilayer structure fabrication.

3.1. Sprayed cell

All five functional layers of a DM-Cell were developed separately by applying thermal spraying technologies. A 48 mm diameter cell with a total functional layer thickness of 180 µm was successfully fabricated. In Fig. 4 a cross-sectional view of the plasma sprayed dual cell deposited on a 1000 µm thick porous metallic substrate can be seen after 250 h of operation at 600°C and 700°C. A 35 µm thick anode layer of BCY and Ni cermet was deposited onto the substrate, and exhibited sufficient porosity after being reduced at high temperature. This was followed by a relatively dense 35 µm thick BCY electrolyte possessing substantially globular and closed porosity, which ensured an acceptable gas leakage rate. A 45 µm thick porous composite DM was deposited on the protonic electrolyte, exhibiting a very good adherence along the radial axis. The second electrolyte YDC was deposited with 50 µm thickness and a relatively higher rate of microcracks, and finally an 10–15 µm LSCF paste was applied which was annealed in situ.
3.2. Tape cast cells

Trilayers of BCY/BCY-porous/BCY were fabricated by tape casting, lamination and successive co-firing. Sintering at 1400°C for 4 h gave almost flat, crack-free, 570 µm thick multilayers. Perfect adhesion of the three laminated layers was also obtained. The central 40 µm thick porous zone is sandwiched between two properly dense layers, as clearly shown in Fig. 5.

Rice starch (RS, Fluka), with average particle size of 6 µm was used as the sacrificial pore-forming agent. The slurries were prepared by adding to the starting ceramic powders the desired amounts of solvent, deflocculant, binder, and plasticiser. The ball milled suspension was de-aerated under vacuum and cast on a moving Mylar carrier (v = 6 mm/s), to obtain, after solvent evaporation, green tapes with the desired thickness.

A uniaxial warm press was used to laminate the green tape layers. The layers with 40 mm diameter were stacked between two polished parallel steel plates, and heated for 15 min at 55°C. During the thermal treatment a constant loading of 7.3 MPa was maintained.

Samples for electrochemical characterisation were electroded with a Pt layer on both sides, realised by screen printing.
4. Results and discussion

A metal supported dual cell was tested in a two-chamber setup for 250 h with 50% H₂–50% N₂ and air at the anode and cathode chamber, respectively. The long-term measurement was interrupted to record several polarisation curves at different temperature and gas supply combinations (Fig. 6a).

In Fig. 6b the maximum power density achieved by the dual cell is presented as 132 mW/cm², 75 mW/cm², and 39 mW/cm² for 800, 700, and 600°C, respectively for hydrogen/oxygen supply. The results show satisfactory gas tightness of the electrolytes, and notable stability in the performance. When compared to intermediate-temperature SOFCs (IT-SOFCs) and protonic ceramic fuel cells (PCFCs) prepared by the same fabrication technology, the DM-Cell resulted in lower performance (by a factor of between 3 and 10), possibly due to the non-optimised microstructure of the porous DM, the most challenging structure for plasma spraying.
Monolithic cells fabricated by tape casting were tested in both SOFC and SOEC mode in a two-chamber setup at different temperatures in the range 600–800°C. In order to highlight the special feature of the DM, a conditioning step was carried out by first polarising the cell in SOFC mode under wet H₂–air feeding for 20 min. This conditioning was then followed by SOEC polarisation with N₂–N₂ (dry or wet–3 at%) feeding. Results for 600°C are reported in Fig. 7, and compared to the same experiment without conditioning, i.e. H₂–air feeding at OCV. A lower resistance was always found after conditioning, with little effect of additional steam injected during polarisation. This result suggests that enhanced water splitting takes place in the DM-Cell due to some form of accumulation of OH⁻ ions, possibly as an adsorbed film at the pores surface in the DM, and confirms the feasibility of a reversible device with the DM-Cell architecture.
5. Conclusions

The DM-Cell is an innovative SOFC concept combining protonic and anionic conductivity, and providing an independent compartment for water formation. Thanks to the IDEAL-Cell project this technology has almost reached protonic ceramic fuel cell standards after only four years of development, and is suitable for fabrication with different industrial processes like plasma spraying and tape casting.

The DM-Cell architecture is particularly favourable for reversible SOFC/SOEC operation.

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